Investigation of Li₂O₂ Film Growth on Carbon Nanotube Electrodes with CeO₂ Nanoparticles

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Unexpected Li_2O_2 film growth with thicknesses of ~60 nm is observed on a 3-D carbon nanotube (CNT) electrode incorporated with cerium dioxide (ceria) nanoparticles (CeO₂ NPs) in Li-O₂ battery. CeO₂/CNT electrodes are collected at different discharge/recharge states from Li-O₂ batteries and investigated by C, Li, and O K-edge XANES. Bulk Li_2O_2 film is decomposed together with formation/decomposition of side products.

Keywords: Li-O₂ battery, XANES, lithium peroxide, Cerium oxide nanoparticles.

<u>背景と研究目的 (Introduction)</u>:

The rechargeable $Li-O_2$ battery has been intensively studied due to its promising potential for providing superior energy density.¹ However, the actual performance with respect to capacity and round-trip efficiency has significantly underperformed at the current stage,¹⁻³ which is closely associated with the nucleation, growth and decomposition processes of Li_2O_2 .

It is generally believed that lithium peroxide (Li₂O₂) electrochemically forms thin films with thicknesses less than 10 nm resulting in capacity restrictions due to limitations in charge transport. However, we observed unexpected Li₂O₂ film growth with thicknesses of up to ~60 nm on a 3-D carbon nanotube (CNT) electrode incorporated with cerium dioxide (ceria) nanoparticles (CeO₂ NPs).⁴ The CeO₂ NPs favor Li₂O₂ surface nucleation owing to their strong binding towards reactive oxygen species (e.g., O₂ and LiO₂). The subsequent film growth results in thicknesses of ~40 nm (a cut-off potential of 2.2 V vs Li/Li⁺), which further increases up to ~ 60 nm with the addition of trace amounts of H₂O which enhances the solution free energy. To understand the growth and decomposition mechanisms of Li₂O₂, it is necessary to analyze the surface and bulk properties of these film-like Li₂O₂. Therefore, CeO₂/CNT electrodes are collected at different discharge/recharge states from Li-O₂ batteries and investigated by C, Li, and O K-edge XANES.

<u>実験 (Experimental)</u>:

i) Synthesis and cell fabrication

CeO₂ NPs were synthesized by thermal decomposition of cerium nitrate precursor in the presence of protecting agent of oleylamine according to the literature.⁵ Around 0.6 mg CeO₂ NPs and CNT (1.8 - 2.0 mg) were well mixed by ultra-sonication in hexane, then made binder-free positive electrode via vacuum filtration. The electrode consisting of 3-D framework of CNTs has many voids (more than 100 nm of size), which is appropriate for O₂ transport in Li-O₂ battery. The positive electrode was annealed at 350 °C in Ar gas and rigorously dried under vacuum, which was then assembled with metallic lithium as the negative electrode, glass fiber separator and tetraethylene glycol dimethyl ether (tetraglyme) with 0.5 M LiClO₄ electrolyte in an Ar-filled glovebox. The complete cell was filled with O₂ gas at atmosphere.

ii) Electrochemical evaluation

Galvanostatic discharge/charge was performed using a battery cycler (WBCS3000, WonATech, Korea) at 25 °C. The applied current (mA/g) is based on the CNT mass. All potentials were referred to Li^+/Li .

iii) Characterization

XANES experiments were performed at the SR center in Ritsumeikan University. C, O and Li K-edge XANES spectra were obtained at BL-02 and BL-11, where all electrode samples and references were loaded in a glove box and directly transferred into the measurement

chamber using a custom-made hermetically sealed vessel.

<u>結果、および、考察 (Results and Discussion)</u>:

In Li-O₂ battery, the nucleation and growth of Li₂O₂ occurs during discharge (DC) through oxygen reduction reaction (ORR, $O_2^* + e^- + Li^+$ \rightarrow LiO₂*, the asterisk mark (*) denotes adsorption on the surface) and a second step via either electrochemical reduction process (LiO₂* $+ e^{-} + Li^{+} \rightarrow Li_2O_2(s))$ or disproportionation $(2\text{LiO}_2(\text{sol}) \rightarrow \text{Li}_2\text{O}_2(\text{s}) + \text{O}_2)$ ^{2, 3} The diffusion of superoxide species (LiO₂*) on electrode surface leads to the predominant surface nucleation and Li₂O₂ film growth (indicated as surface-mediated growth). Morphology of the fully discharged Li₂O₂ product formed on CeO₂/CNT electrode is characterized by SEM and TEM (Figure 1). A thick film-like Li_2O_2 (thickness ~ 40 nm) is observed on CeO2/CNT electrode when using the dry electrolyte (0.5 M LiClO₄ in tetraglyme, with H_2O content < 20 ppm). With the presence of 2000 ppm of H₂O in the electrolyte, the film thickness increased to ~60 nm. Results demonstrate the contribution of increased LiO₂(sol).



Figure 1. SEM and TEM (inset) images of CeO_2/CNT electrode surfaces after full discharge (DC) with (a) dry tetraglyme and (b) 2000 ppm of H₂O. The white dashed line in TEM image is a guide to indicate the CNT surface.

Galvonostatic discharge/charge profiles with a limited discharge capacity of 2 mAh is shown in Figure 2. The discharge of CeO₂/CNT electrode in Li–O₂ cell shows a stable potential plateaus at \sim 2.63 V (vs. Li/Li⁺), whereas recharge of Li₂O₂ thick films required a high overpotential (over

4.0V). CeO₂/CNT electrodes at 1DC (100% discharge, 2.0 mAh), 0.25RC (25% recharge) 0.5RC and 1RC are collected from Li-O₂ cells and characterized by O K-edge.



Figure 2. Galvanostatic discharge (DC)/recharge (RC) curves at a current density of 0.1 mA with a limited DC capacity of 2.0 mAh using CeO₂/CNT electrode as cathode in Li-O₂ battery. The water content in the electrolyte is < 20 ppm.



Figure 3. O K-edge XANES spectra of soaked (as prepared and soaked in the tetraglyme electrolyte), 1DC (Q = 2.0 mAh), 0.25RC, 0.5RC and 1RC CeO₂/CNT electrodes with (a) surface-sensitive total electron yield (TEY) mode and (b) bulk-sensitive partial fluorescence yield (PFY) mode. The reference spectra are Li₂O₂, HCO₂Li, CH₃CO₂Li and Li₂CO₃. The galvanostatic conditions are shown in Figure 2.

O K-edge XANES spectra of as-prepared CeO₂/CNT electrode exhibits three main peaks related to O2*p*-Ce4*f* (529.5 eV), O2*p*-Ce5*d*- e_g (532.2 eV) and O2*p*-Ce5*d*- t_{2g} (536.6 eV) hybridization in CeO₂ NPs (Figure 3).⁶ After 1DC, σ^* (O–O) absorption peak of Li₂O₂ is predominant at ~530 eV,^{7.8} which is partially diminished at 0.25RC and 0.5RC suggesting Li₂O₂ decomposition, as demonstrated by appearance of CeO₂-related peaks such as the O2*p*-Ce5*d*- e_g and O2*p*-Ce4*f*, and completely

disappear after 1RC. Along with the primary product of Li₂O₂, side products are also observed. The 1DC spectrum with the TEY mode reveals a shoulder corresponding to lithium carboxylates (HCO₂Li and CH₃COOLi) at 531.7-532 eV. After 0.25RC, an absorption peak at ~532 eV is significant, which is attributed to both lithium carboxylates and CeO_2 (O2p-Ce5d-e_g). After carboxylates 0.5RC. lithium are more pronounced than the Li₂O₂ being decomposed. Besides, there is an additional small shoulder related to π^* (C=O) absorption of lithium carbonate (Li₂CO₃) appears at 532.8 eV⁸ with the TEY mode. These side products result from degradation of electrolyte solution and CNT.^{9,10} After 1RC, the side product peaks disappear almost completely and the three main peaks from CeO₂ are visible.

In summary, we investigated the electrochemical performance of Li-O_2 cells with CeO_2 NPs decorated on the CNT. An unexpected film product is observed on the discharged electrode, with thickness of ~ 40 nm, and increased to ~ 60 nm when enhancing the solvation of LiO_2 by addition of H_2O in the electrolyte. Through the analysis of O K-edge XANES spectra, the discharge product is mainly Li_2O_2 , with trace amount of carbonaceous side product on the surface. The recharge process is accompanied with further formation and decomposition of side products, which are mainly lithium carbonate and lithium carboxylates.

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<u>論文・学会等発表(予定)(Articles and</u> <u>meetings)</u>

<u>Articles</u>

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Meetings

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