

S25002

リチウム空気電池の放電生成物の構造解析と副生成物の定量分析

Structure analysis and byproduct quantification of discharge products in Li-O₂ batteryKeisuke Mukai^a Do Duy Khiem,^a^a National Institute for Fusion Science, National Institute for Fusion Science, 322-6 Oroshi-cho, Toki, Gifu 509-5292, Japane-mail: mukai.keisuke@nifs.ac.jp**Abstract**

Discharged products in lithium air batteries at different (dis)charging states were examined to elucidate their forming/decomposing reaction mechanism. X-ray absorption fine structure (XAFS) spectra of partially/fully (dis)charged samples on the carbon paper cathode were obtained at SR center BL-11 at Ristumeikan University. XAFS spectra showed that Li₂O₂ are electrochemically formed with byproducts of LiRCO₂ and Li₂CO₃ during discharging. PEY and PFY spectra showed that these byproducts formed not only near-surface but also inside of the discharge product toroid. By LCA analysis on the PEY spectra from the carbon paper cathode after fully-charged in fluorinated amide electrolyte (DETFA-50), less byproducts remained on the cathode surface in consistent our SEM/EDX observation. This result suggests that decomposition of the discharged products including insulating byproducts are affected by additives in amide electrolytes.

Keywords: Lithium, Li K-XANES, O K-XANES, byproduct**Background**

Li-O₂ battery is a promising next-generation energy storage systems because of its high theoretical energy density. Its application is hindered by low cycle performances, which possibly originates from formation of less-decomposable discharge products, including insulating carbonate byproducts. To understand the reaction mechanism, the chemical phases of Li-related products formed during operation in different electrolytes must be clarified. In this study, we have employed XAFS to identify and quantify phases formed during (dis)charging.

Experiment

Cathode samples were prepared in the electrochemical cell. Li foil (25 mm diameter and 400 μm thickness). Carbon paper (CP, 20 mm diameter, TGP-H-060, Toray) was used as a positive electrode. The CP was prepared by immersing pristine CP in hot acid (H₂SO₃:HNO₃ = 3:1) for 2 h. A glass-fiber filter (GF/A, 22 mm diameter, Whatman) was used as the separator after it was dried under a vacuum at 110 °C overnight. Dehydrated DMA (Wako) was used as the solvent for the electrolyte. LiNO₃ (Kishida, Japan) was used as the salt. The electrolytes were prepared by stirring the salt and solvent overnight. Discharge products were formed on the CP in three different electrolytes; *N,N*-dimethylacetamide (DMA) and DETFA-50 (50 vol.% *N,N*-diethyl-2,2,2- trifluoroacetamide (DETFA) + 50 vol.% DMA). The electrochemical cell employed in this study is previously described elsewhere [1].

Results and Discussion :

Figures 1 shows the XAFS spectra of O in DMA and DETFA. XAFS O K-edge spectra acquired peaks at approximately 530 eV for all samples, corresponding to the σ*(O–O) absorption peak of Li₂O₂. In addition to Li₂O₂, LiRCO₂ and Li₂CO₃ peaks were observed at ~532 and ~533 eV as byproducts. These phases are consistent with our previous analysis using Fourier transform infrared spectrometer (FT-IR).

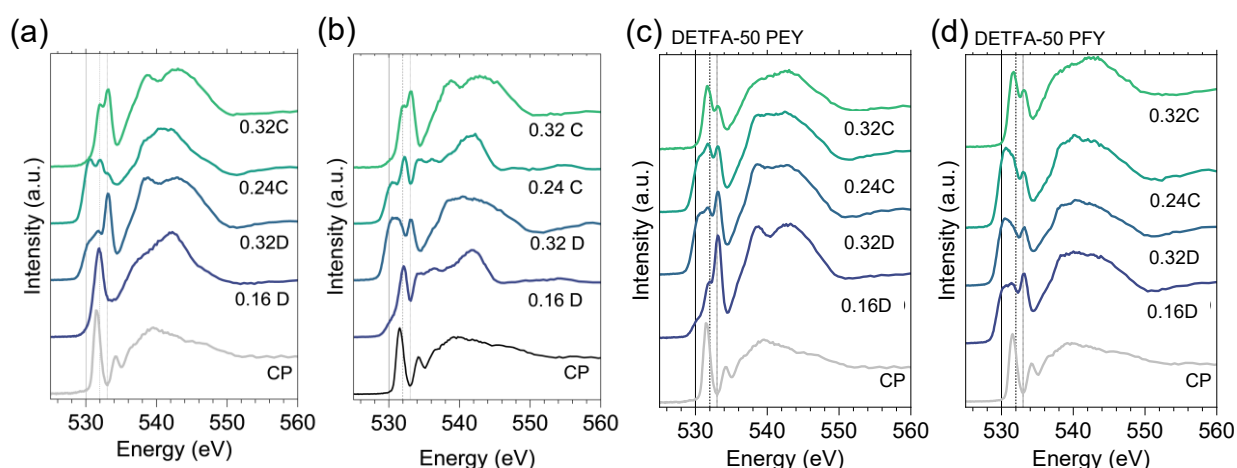


Fig. 1. Spectra of O K-edge XAFS at different (dis)charging states in (a) DMA by PEY, (b) DMA by PFY, (c) DETFA-50 PEY, and (d) DETFA-50 by PFY. The samples were (dis)charged to 0.16, 0.24, and 0.32 mAh cm⁻² at a constant current of 0.1 mA cm⁻². D and C denote discharge and charge.

Fig. 2 shows the results of linear combination analysis (LCA) for the CP samples discharged to 0.32 mAh cm⁻² and subsequently charged to 0.32 mAh cm⁻² at a constant current of 0.1 mA cm⁻². The LCA was performed using reference spectra from CP and standard powders of Li₂CO₃, CH₃COOLi, and CF₃COOLi. The fittings for the DMA and DETFA-50 samples resulted in R-factors of 0.0216 and 0.0218, respectively. In the CP sample fully charged in DMA, the spectrum was composed mainly by the byproducts, where the byproduct proportions of ~90% in DMA. On the contrary, the remaining byproducts in the CP sample fully-charged in DETFA-50 was ~50%. This indicates that the insulating byproducts remained on the CP surface even after the full charging, while Li₂O₂ were electrochemically decomposed. Notably, the remaining byproducts were less in DETFA-50 electrolyte. These results are fully consistent with our SEM/EDS and FT-IR results, in which sub-micron byproduct particles remained on the CP after being fully charged in DMA. These results may imply facilitated decomposability of discharge products in DETFA-50. Effect of fluorinated electrolytes on decomposability and possible link to structures of discharge products will be further investigated in our future study.

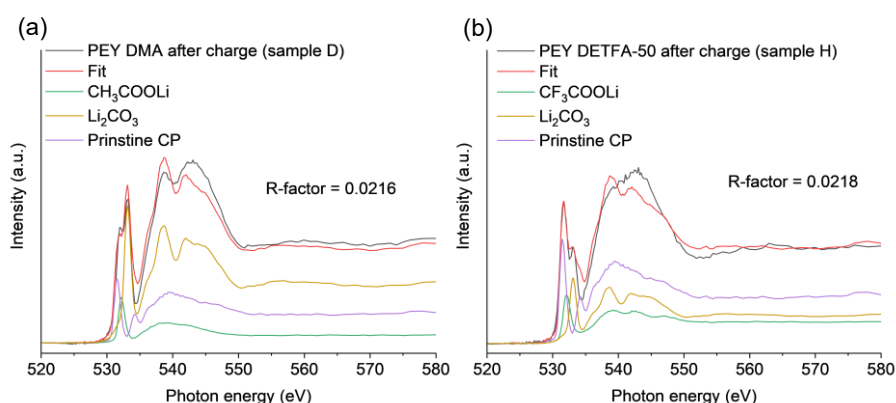


Fig. 2. Fittings by LCA on the PEY spectra from cathode samples fully-charged in (a) DMA and (b) DETFA-50.

References

- [1] Morimoto, K.; Kusumoto, T.; Nishioka, K.; Kamiya, K.; Mukouyama, Y.; Nakanishi, S. Dynamic Changes in Charge Transfer Resistances during Cycling of Aprotic Li–O₂ Batteries. *ACS Appl. Mater. Interfaces* 2020, 12 (38), 42803–42810.